

Electronic Supporting Information

A Closer Look at the Development and Performance of Organic/Inorganic Membranes Using 2,4,6-tris[3(triethoxysilyl)-1-propoxyl]-1,3,5-triazine (TTESPT)

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Electronic Supporting Information-1 (ESI-1)

1.1 Synthesis of 2,4,6-tris[3(triethoxysilyl)-1-propoxyl]-1,3,5-triazine (TTESPT)

TTESPT was synthesized in a (30 mL) two-necked flask equipped with a stirrer and a reflux condenser, into which was placed (4.40 g, 17.7 mmol) of 2,4,6-tris(allyloxy)-1,3,5-triazine and (10.40 g, 63.3 mmol) of triethoxysilane in (10 mL) of toluene. Two drops of (0.1 M) H_2PtCl_6 in *i*PrOH were added, and the mixture was heated to reflux for 4 hours. Excess reactant and the solvent were removed under vacuum to give a colorless liquid (12.08 g). All reactions were performed under an atmosphere of dry nitrogen. Figure S1 is a schematic diagram of the synthesis of 2,4,6-tris[3(triethoxysilyl)-1-propoxyl]-1,3,5-triazine (TTESPT)

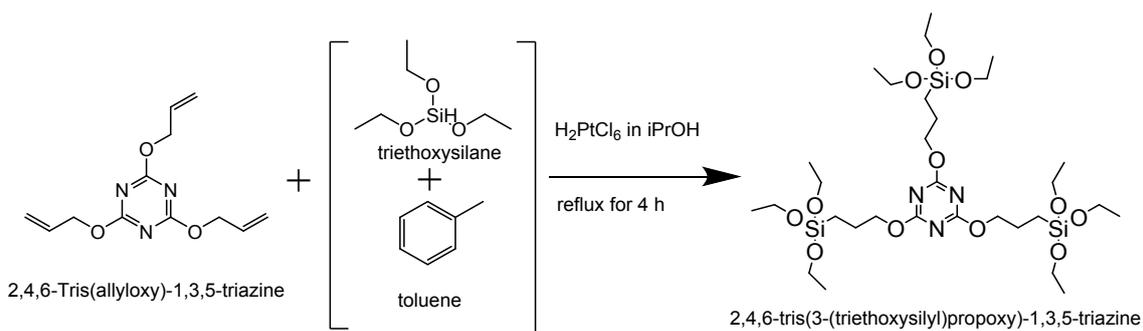
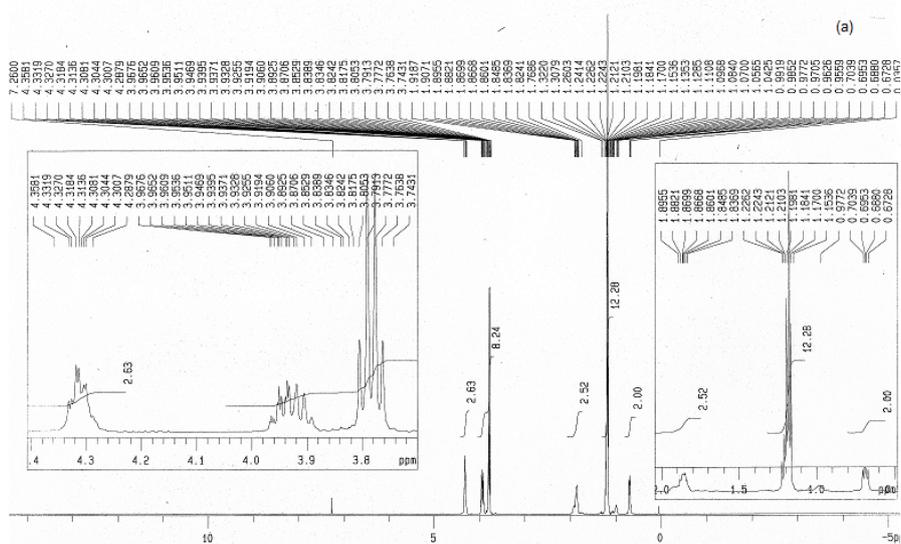


Figure S1 Schematic diagram of the synthesis of 2,4,6-tris[3(triethoxysilyl)-1-propoxyl]-1,3,5-triazine (TTESPT)

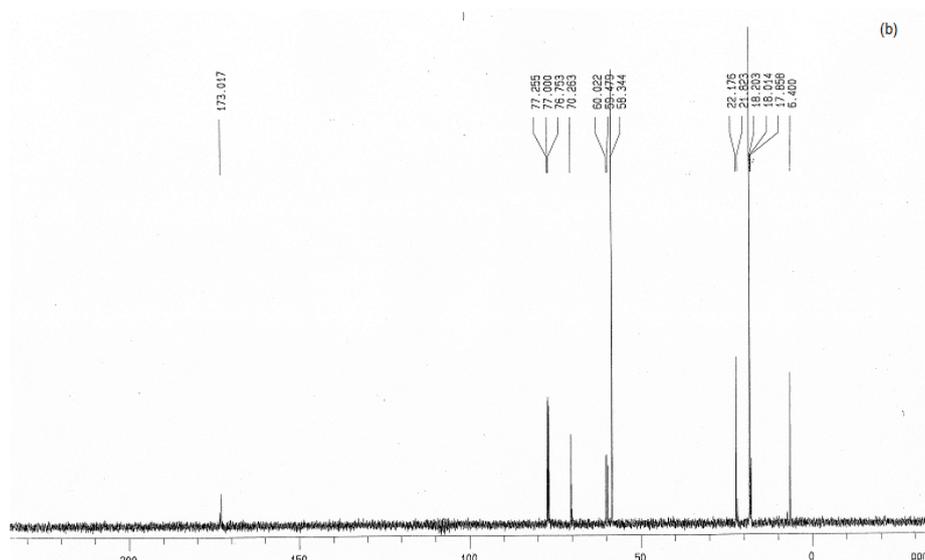
Anal. Calcd and NMR spectra for $C_{30}H_{63}N_3O_{12}Si_3$ (TTESPT)^{1a,b} are shown as below (Figure S2):

¹H NMR (δ $CDCl_3$) 0.69 (m, 6H, CH_2Si) 1.18 (t, 27H, CH_3 , $J = 7.0$ Hz), 1.82-1.92 (m, 6H, CH_2), 3.78 (q, 18H, CH_2O , $J = 7.0$ Hz), 4.28-4.36 (m, 6H, CH_2O);

¹³C NMR (δ $CDCl_3$) 6.40 (CH_2Si), 18.20 (CH_3), 22.18 (CH_2), 58.34 (OCH_2), 70.26 (OCH_2), 173.02 (triazine carbon).



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32

33 **Figure S2** NMR spectra of 2,4,6-tris[3(triethoxysilyl)-1-propyl]-1,3,5-triazine $C_{30}H_{63}N_3O_{12}Si_3$ (TTESPT) solution

34 (a) 1H NMR and (b) ^{13}C NMR

35

36 **1.2. Preparation of a TTESPT-silica derived membrane**

37 A TTESPT-derived silica solution was prepared using hydrochloric acid as a catalyst. First, TTESPT as a silica
 38 precursor was homogeneously dissolved in an ethanol solution. A mixture of water and hydrochloric acid were then
 39 added dropwise to the solution under vigorous stirring, resulting in a final solution with molar ratios of
 40 TTESPT/ H_2O /acid = (1/240/0.2). Then, the solution was kept in a closed system under continuous stirring at 25 °C
 41 for 6 h to form silica sols. TEOS, and BTESE-derived silica solutions were prepared for comparison, and the details
 42 of the preparation procedure can be found elsewhere.²⁻⁴ Porous α -alumina tubes (porosity: 50%, average pore size: 1
 43 μm , outside diameter: 10 mm) were used as the supports for TTESPT-derived silica membranes. α -Alumina particles
 44 (average particle diameter: 0.2, 1.9 μm) were coated on the outer surface of a porous support using silica-zirconia

45 colloidal sol as the binder, and the support was fired at 550 °C for 30 min to make the surface smooth. These
46 procedures were repeated several times to cover large pores that might have resulted in pinholes in the final
47 membrane. Then, a SiO₂-ZrO₂ (Si/Zr = 1/1) solution diluted to about (0.5 wt%) was coated on the substrate to form
48 an intermediate layer with pore sizes of several nm, followed by calcination at 550 °C for about 30 min. Finally, the
49 TTESPT-derived silica layer was fabricated by coating a TTESPT solution, followed by drying and calcination at
50 200, 300 and 400 °C under nitrogen for 30 min.

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52 1.3. Characterization of TTESPT-silica derived membrane

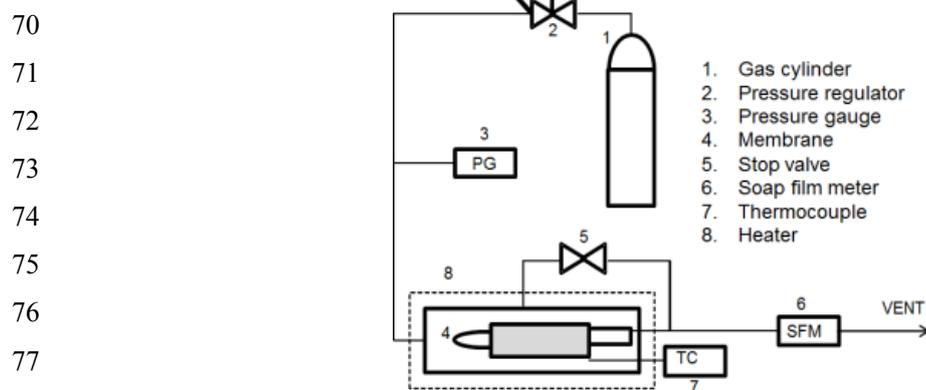
53 ¹H NMR and ¹³C NMR spectra were recorded using a JNM-LA500 spectrometer. Thermogravimetric analysis (TG)
54 was performed to investigate the decomposition behavior of the organic groups in the silica matrix at temperatures
55 ranging from 100 to 800 °C with a heating rate of 10 °C min⁻¹ in a Helium flow of 300 ml min⁻¹ using a
56 Thermogravimetric Mass Spectrometer (TGA-DTA-PIMS 410/S, Rigaku, Japan). The TTESPT-derived film coated
57 on the KBr plate was fired between 200 and 550 °C under an N₂ atmosphere, and was characterized by in situ Fourier
58 transform infrared (FTIR) spectroscopy (FTIR-JASCO, Japan). The morphology of TTESPT membranes were
59 examined by using a Hitachi S-48000 scanning electron microscope (SEM). A N₂ adsorption isotherm was carried
60 out at 77 K to study the micropore structures of the silica gels. Prior to the measurement, all the silica gel samples
61 were evacuated at 150 °C for 12 h. The measurements were carried out using BELMAX (BEL JAPAN INC., Japan).

62

63 1.4. Single gas permeation measurement

64 Gas permeation tests were performed at temperatures ranging from 200 to 50 °C using a single component of He, H₂,
65 CO₂, N₂, CH₄, C₃H₈, and SF₆ before and after the coating process as shown in Figure S3. The permeation stream was
66 maintained at atmospheric pressure, and the pressure drop through the membranes was maintained at 1 bar. Prior to
67 the measurement, all membranes were outgassed in a He flow of 50 ml min⁻¹ at 200 °C for 8 to 12 h to remove any
68 water that possibly absorbed through the membranes.

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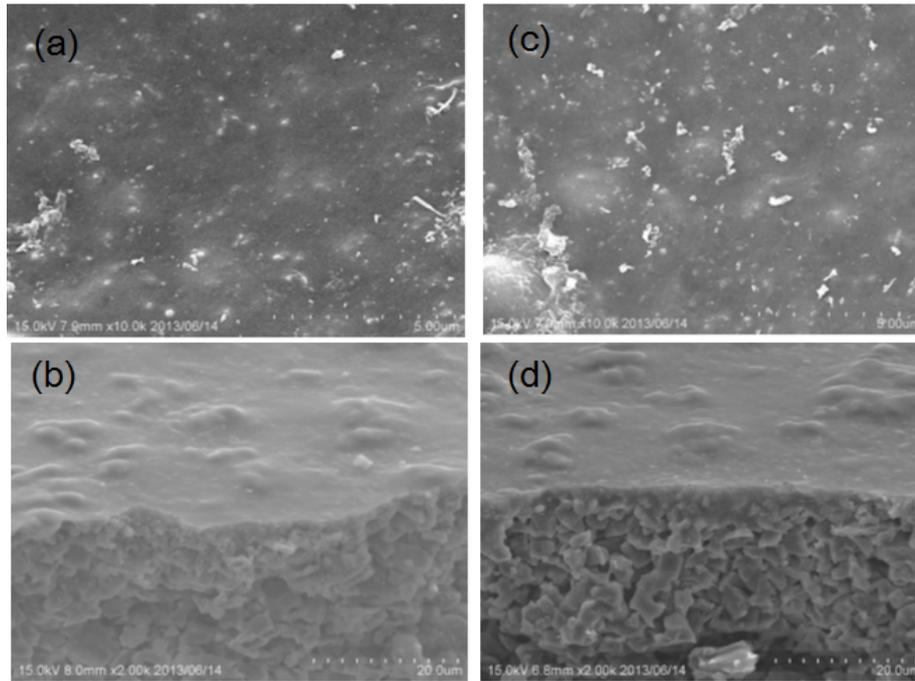


79 **Figure S3** Schematic diagram of the experimental apparatus for single gas permeation measurement

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81 **Electronic Supporting Information-2 (ESI-2)**

82 Figure S4 shows the SEM images of the surfaces, (a) and (c), and cross-sections, (b) and (d), of TTESPT membranes
83 calcined at 200 and 400 °C under N₂ environment at magnification of 10,000 and 2,000, respectively. As shown in
84 Figures 4 (a) and (c), almost no change occurred to the membrane top surface areas. On the other hand, a very thin
85 layer of TTESPT was successfully coated on the top of SiO₂-ZrO₂ intermediate–alumina particle layer from the cross
86 section shown in Figures 4 (b) and (d).



87

88 **Figure S4** SEM images of the surface and cross sections of TTESPT membranes calcined at 200 °C (a and b); 400 °C
89 (c and d) under N₂ environment

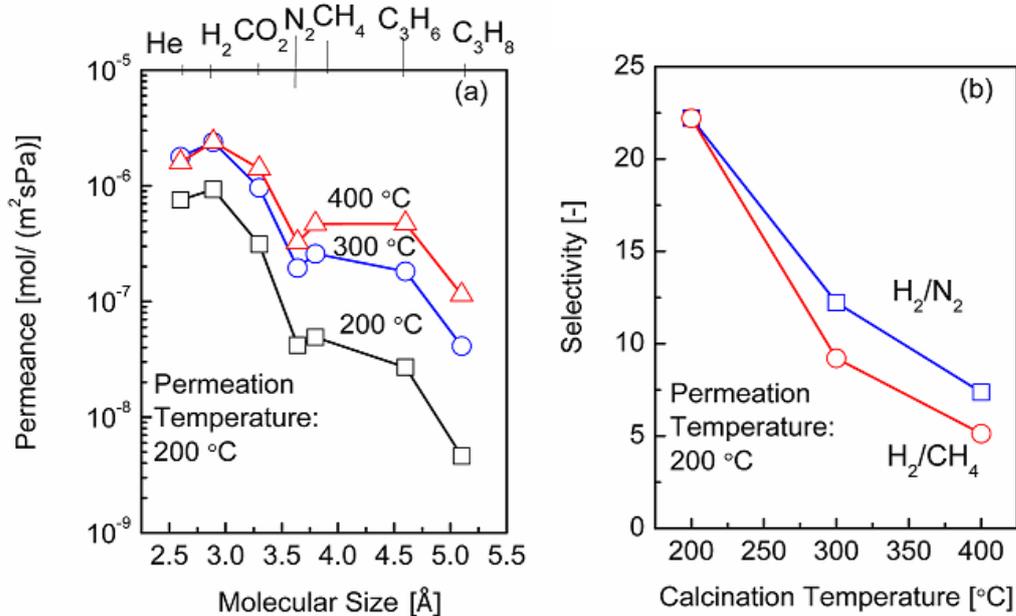
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91 **Electronic Supporting Information-3 (ESI-3)**

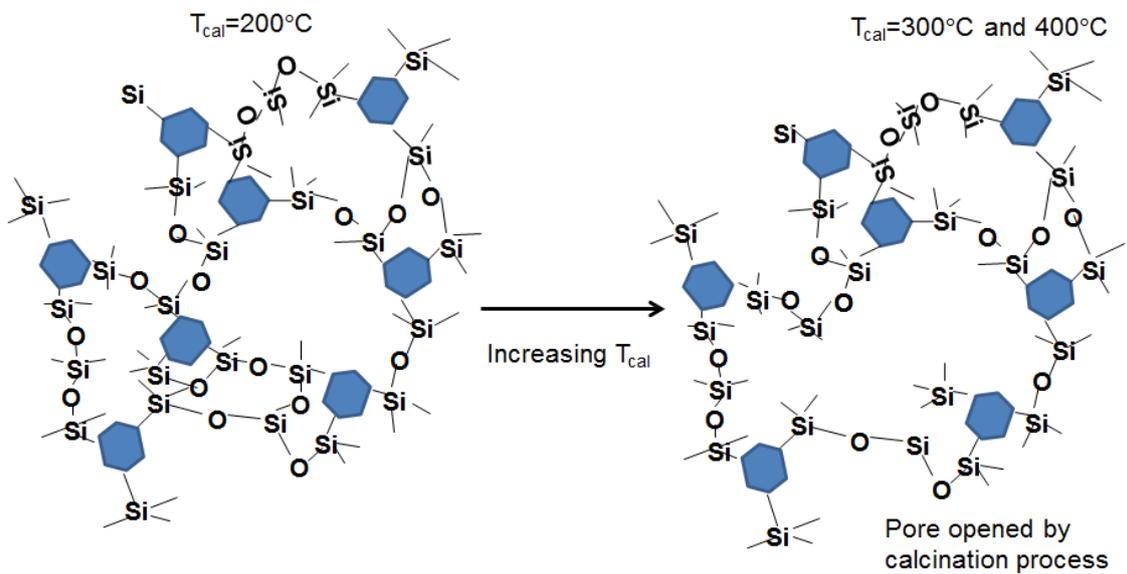
92 **2.1 Relationship between temperature dependence of single-gas permeances and quantitative pore size**
93 **measurement (template technique)**

94 Figure S5 shows the gas permeance and selectivity properties for TTESPT-derived silica membranes calcined at 200,
95 300 and 400 °C at a permeation temperature of 200 °C. As shown, as the calcination temperature was increased from
96 200 to 400 °C, and the H₂ permeance increased from 0.93×10⁻⁶ to 2.39×10⁻⁶ mol/(m² s Pa), while the H₂/N₂ and
97 H₂/CH₄ selectivity decreased from 22 to 7 and 22 to 5, respectively. Calcination at high temperatures clearly showed
98 a different tendency from that observed for the TEOS-derived silica membrane. deVos *et al.*⁵⁻⁶ reported a decrease in
99 H₂ permeance from 2 ×10⁻⁶ to 5× 10⁻⁷ mol/(m² s Pa) by calcined TEOS-derived silica membranes at 400
100 to 600 °C, and an increase in H₂/CH₄ selectivity of from 560 to 4,000. The change in the gas permeation properties of
101 TEOS membranes at different calcination temperatures might have resulted from densification, which caused a
102 decrease in the amount of terminal hydroxyl groups at the internal surface of the silica.⁵⁻⁶ However, in our case, an

103 opposite phenomenon was seen. This phenomenon was known as the template method. Figure S6 shows the
 104 schematic images of the formation of TTESPT-silica pore networks via the template method. Obviously, this method
 105 can be used in order to tune the pore size of TTESPT-silica derived membrane.



106
 107 **Figure S5** (a) Gas permeance and (b) selectivity properties for TTESPT-derived silica membranes calcined at
 108 different temperatures of 200, 300 and 400 °C under a N₂ atmosphere at permeation temperature 200 °C
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110
 111 **Figure S6** Schematic structure of a pore network of TTESPT via the template method.
 112
 113
 114 In order to quantitatively evaluate the pore size of these membranes, a normalized Knudsen-based permeance (NKP)

115 method was applied.⁷ This NKP method is based on the modified gas translation (GT) model that was derived by
 116 modifying the original GT model proposed by Xiao and Wei⁸ and by Shelekhin et al.⁹ for the determination of
 117 membrane pore sizes of less than 1 nm. As mentioned by Lee et al.,⁹ NKP is the permeance ratio of the *i*th component
 118 (P_i) to that predicted from the *j*th component based on the Knudsen diffusion mechanism, $P_j = \sqrt{M_j/M_i}$ where
 119 helium, the smallest molecule, is used as a reference (*j*th component). NKP can be expressed by Eq. (1).

$$120 \quad NKP = \frac{P_i}{P_{He} \sqrt{M_{He}/M_i}} \quad (1)$$

121 NKP is equal to 1 for all gases; Knudsen diffusion is the dominant mechanism. In the original GT as shown in Eq. (2),
 122 the molecular size is not explicitly incorporated based on the assumption that the size of the gas is small enough.

$$123 \quad P_i = d_p \rho_{g,i} \sqrt{\frac{8}{\pi M_i RT}} \frac{\varepsilon_i}{\tau_i L_i} \exp\left(-\frac{E_{p,i}}{RT}\right) \quad (2)$$

124 The diffusion distance of the *i*th component, $d_{p,i}$ is assumed to be $(d_p - d_{k,i})$ for the *i*th component (molecular size: $d_{k,i}$)
 125 due to the steric hindrance, the permeance, P_i , through microporous inorganic membranes (membrane porosity, ε ;
 126 pore diameter, d_p ; tortuosity, τ , thickness, L) can be formulated as the modified GT-model in Eq. (3) by using the
 127 molecular weight of the permeating component, M , gas constant, R , and the temperature, T . The probability of the *i*th
 128 component, $\rho_{g,i}$, corresponds to the geometrical probability; that is, the product of the effective area for permeation,

$$129 \quad (d_p - d_{k,i})^2 / d_p^2 \quad (3)$$

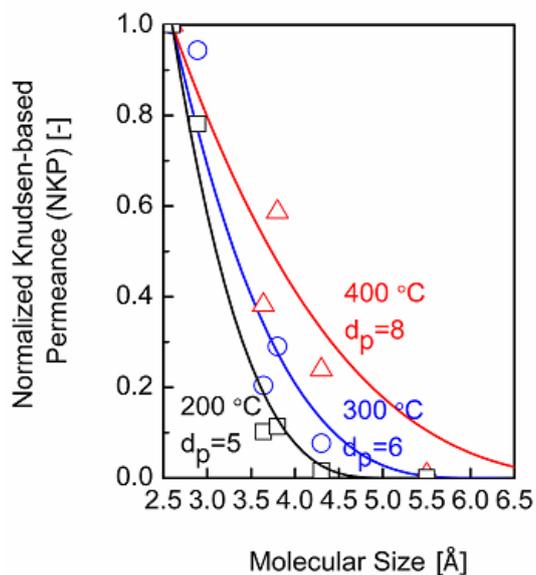
$$130 \quad P_i = \frac{(d_p - d_{k,i})}{3} \left(\frac{(d_p - d_{k,i})^2}{d_p^2} \right) \sqrt{\frac{8}{\pi M_i RT}} \frac{\varepsilon_i}{\tau_i L_i} \exp\left(\frac{E_{p,i}}{RT}\right) \quad (3)$$

131 Therefore, by inserting Eq. (3) into Eq. (1), and an assumption of the negligible difference of the activation energy of
 132 permeation, NKP can be formulated as Eq. (4).

$$133 \quad NKP = \frac{(d_p - d_{k,i})^3}{(d_p - d_{k,He})^3} \quad (4)$$

134 The pore size of TTESPT-derived silica membranes obtained via the NKP method also shifted from 0.5 to 0.8 nm
 135 when the calcination temperature was increased from 200 to 400 °C (Figure S7). This can be ascribed to the partial
 136 removal of triazine-based linking units as confirmed by FTIR spectra. A similar trend has also been reported for
 137 MTES, TEDMDS and MTES and TEOS-derived silica membranes.^{4,7} An enhancement of gas permeance was
 138 observed after the membranes were calcined at a higher temperature under environments of N₂ and air.

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140

141 **Figure S7** Normalized Knudsen-based permeance as a function of molecular size at a permeation temperature of
 142 200 °C; points are experimental and curves are calculated for TTESPT-derived silica membranes calcined at different
 143 temperatures of 200, 300 and 400 °C.

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145 **References**

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