

## Supporting Information

### Surface-Coordination-Induced Selective Synthesis of Cubic and Orthorhombic NaNbO<sub>3</sub> and Their Photocatalytic Properties

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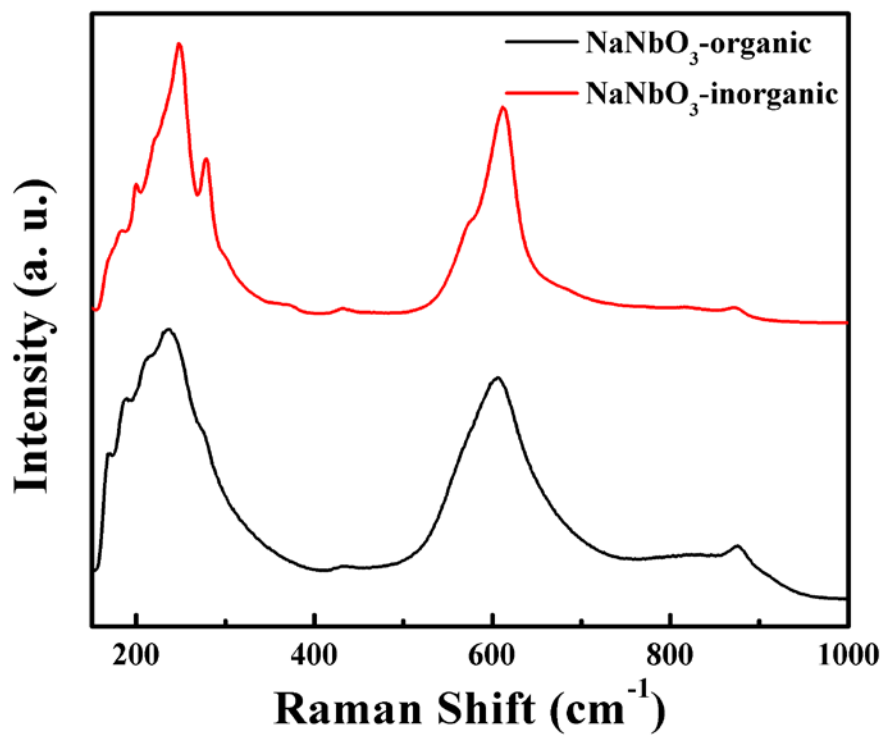
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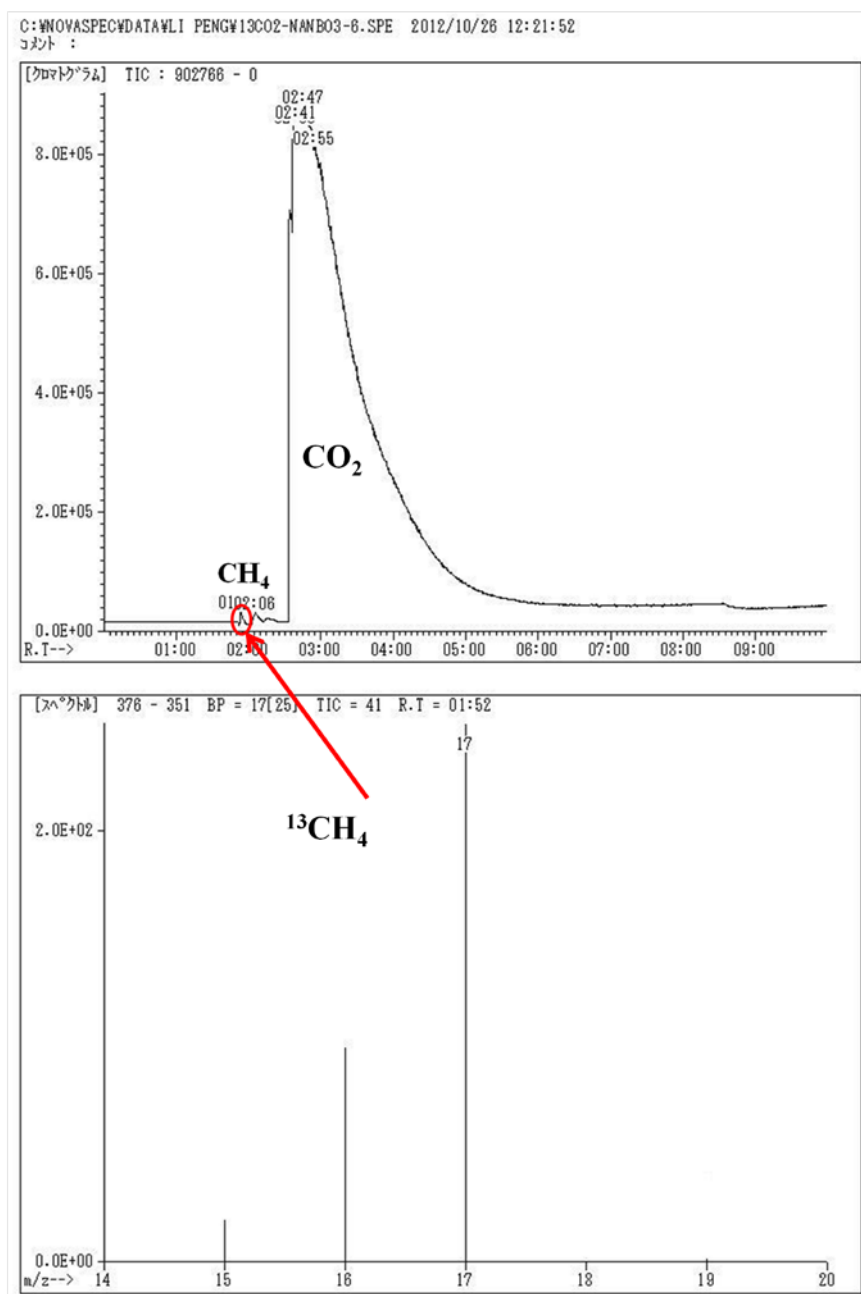
Electronic mail: [jinhua.ye@nims.go.jp](mailto:jinhua.ye@nims.go.jp).

### SI-1 Raman spectra of NaNbO<sub>3</sub>-organic and NaNbO<sub>3</sub>-inorganic.



**Fig. S1** Raman spectra of NaNbO<sub>3</sub>-organic and NaNbO<sub>3</sub>-inorganic.

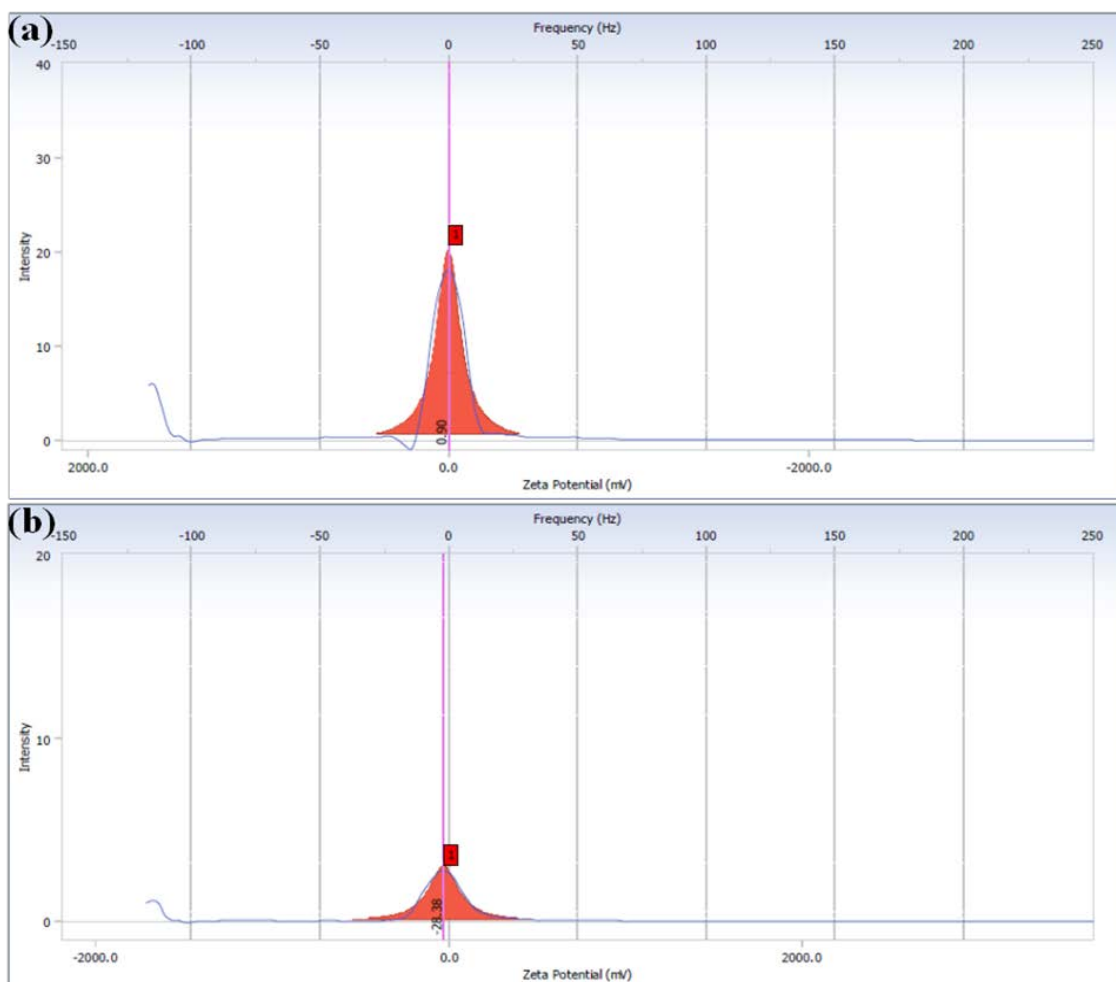
## SI-2 The reference experiment using $^{13}\text{CO}_2$ as carbon source in photocatalytic $\text{CO}_2$ reduction over $\text{NaNbO}_3$ -organic.



**Fig. S2** GC-MS spectra of the products of photocatalytic  $^{13}\text{CO}_2$  reduction over  $\text{NaNbO}_3$ -organic after 16 h irradiation.

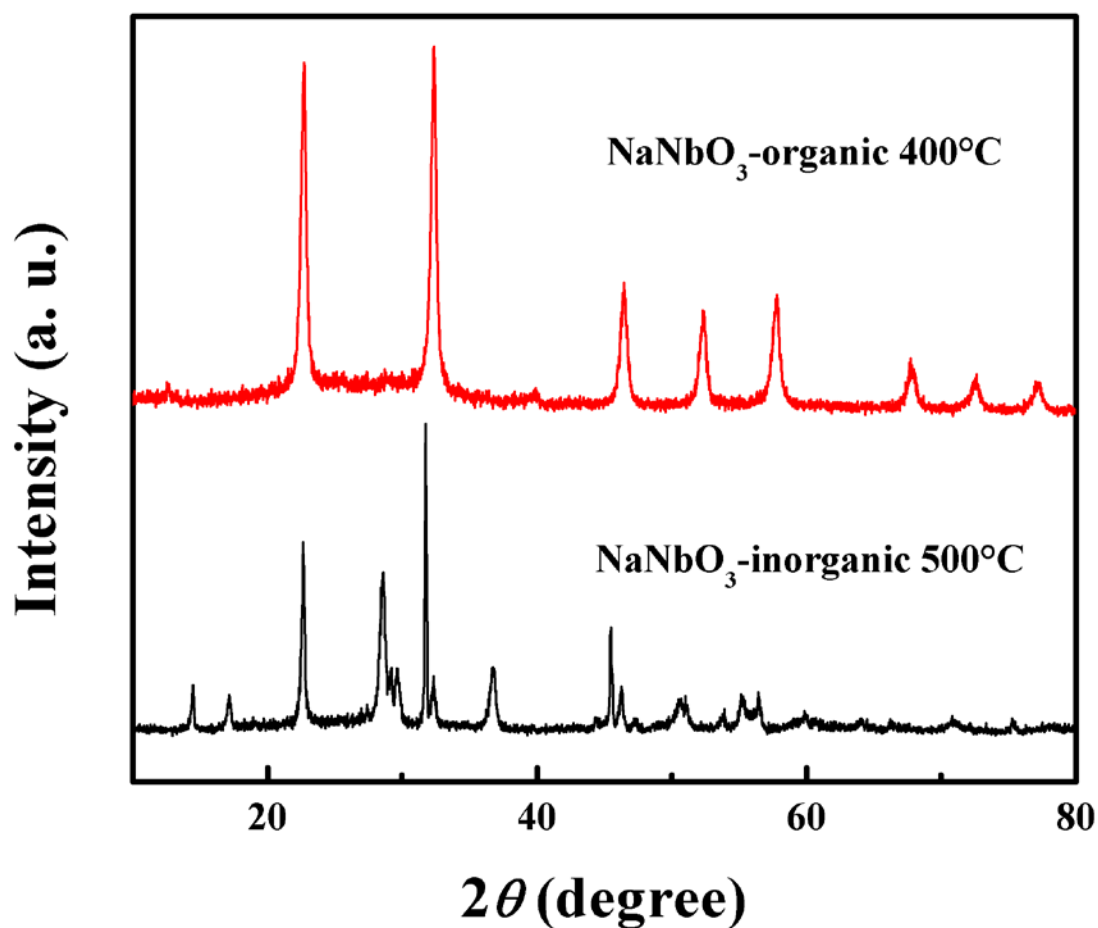
*In the GC spectrum, the main peak with the retention time 02:50 is confirmed to be CO<sub>2</sub>. Another peak with the retention time 01:52 is the main organic product CH<sub>4</sub>. As observed from the mass spectrum, the base peak with m/z of 17 could be determined to be the parent peak (M<sup>+</sup>) of <sup>13</sup>CH<sub>4</sub>. The peaks with m/z of 16 and 15 could be recognized as the fragment peaks M-1<sup>+</sup> and M-2<sup>+</sup>. The intensity ratio of M<sup>+</sup>, M-1<sup>+</sup> and M-2<sup>+</sup> matches that of <sup>12</sup>CH<sub>4</sub>. Therefore, the main organic product <sup>13</sup>CH<sub>4</sub> is verified to be generated from the reduction of <sup>13</sup>CO<sub>2</sub>.*

### SI-3 The mobility distributions of $\text{NaNbO}_3$ -inorganic and $\text{NaNbO}_3$ -organic precursor colloidal particles.



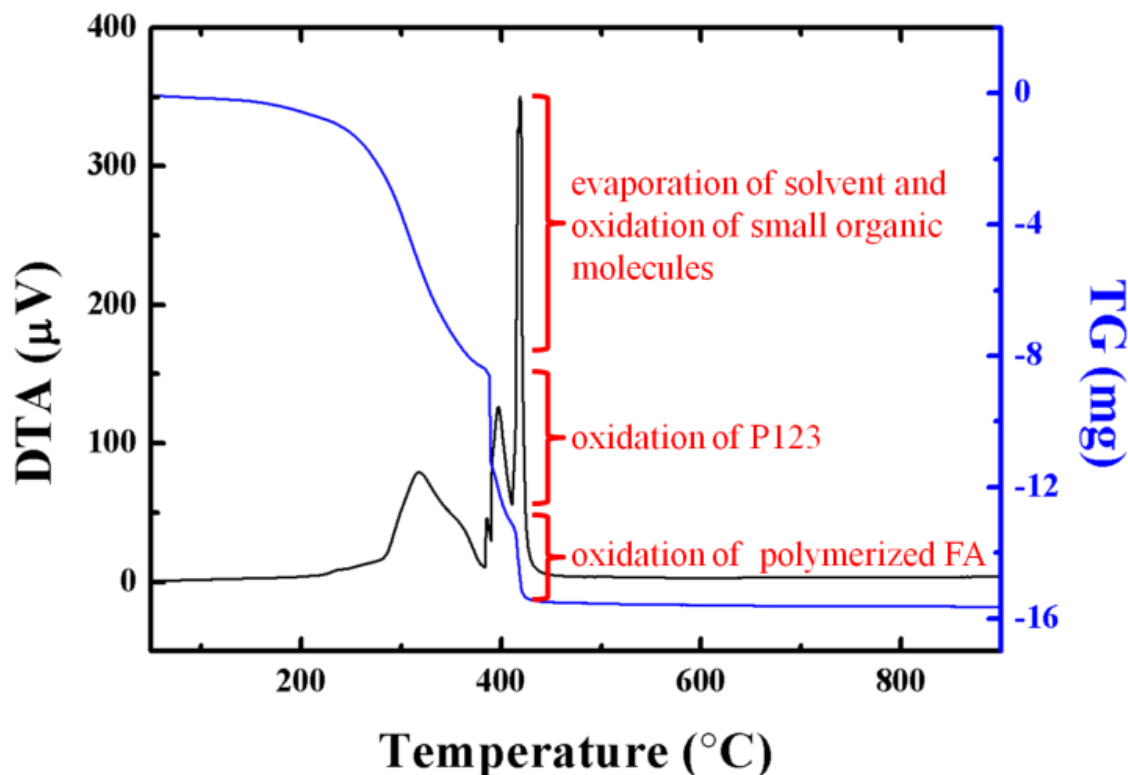
**Fig. S3** (a) The mobility distributions of  $\text{NaNbO}_3$ -inorganic precursor colloidal particles. (b) The mobility distributions of  $\text{NaNbO}_3$ -organic precursor colloidal particles.

**SI-4 The XRD patterns of the low temperature oxidation samples of  $\text{NaNbO}_3$ -inorganic and  $\text{NaNbO}_3$ -organic.**



**Fig. S4** The XRD patterns of  $\text{NaNbO}_3$ -organic sample oxidized at 400°C and  $\text{NaNbO}_3$ -inorganic sample oxidized at 500°C.

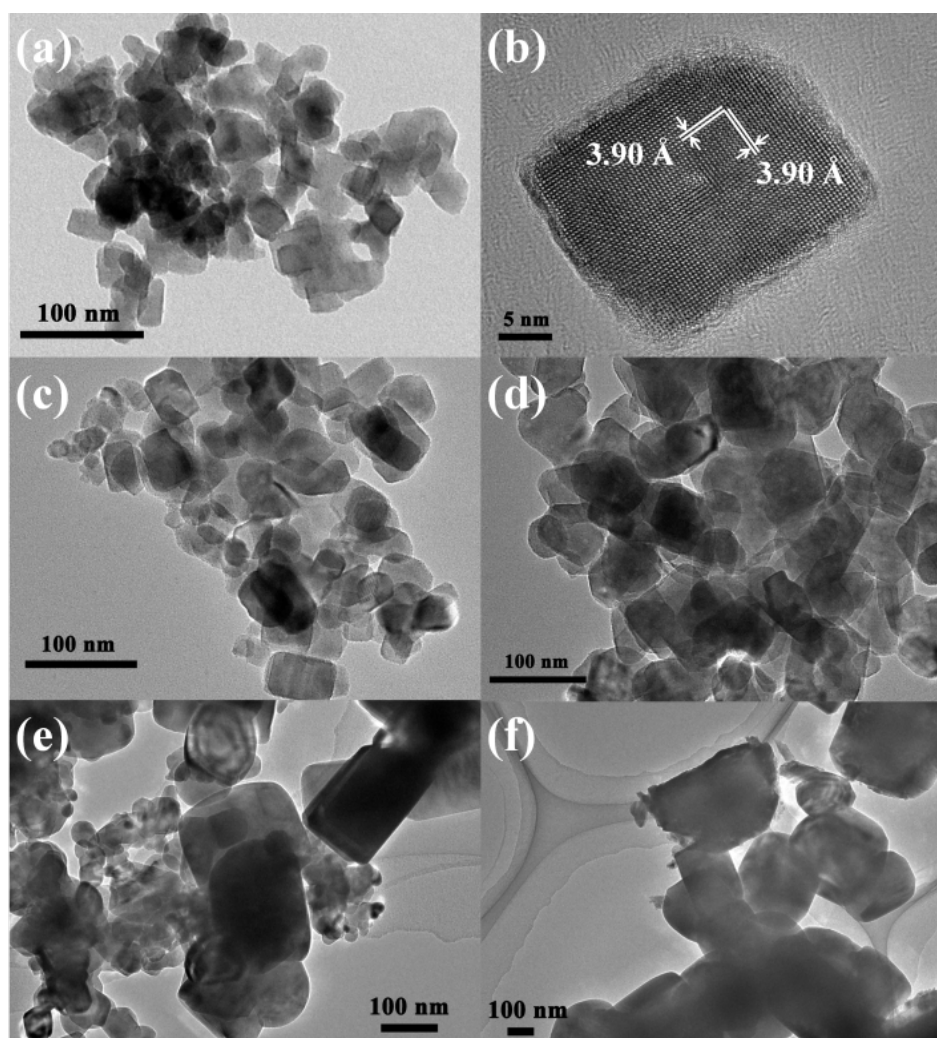
### SI-5 TG-DTA plots of the $\text{NaNbO}_3$ -organic polymer precursor.



**Fig. S5** TG-DTA plots of the  $\text{NaNbO}_3$ -organic polymer precursor.

For the polyfurfuryl alcohol precursor with P123, a subsequent TGA runs in air with a flow rate of  $20 \text{ mL} \cdot \text{min}^{-1}$  shows a significant weight loss that occurs between 200 and 385°C from the evaporation of the remaining solvent and the oxidation of low molecular weight polyfurfuryl alcohol to  $\text{CO}_2$ . As expected, there is another rapid weight loss between 385 and 415°C from the oxidation of P123. The third weight loss from 415 to 435°C was attributed to the decomposition of high molecular weight polyfurfuryl alcohol.

## SI-6 The TEM images of the oxidation temperature relative NaNbO<sub>3</sub> samples.



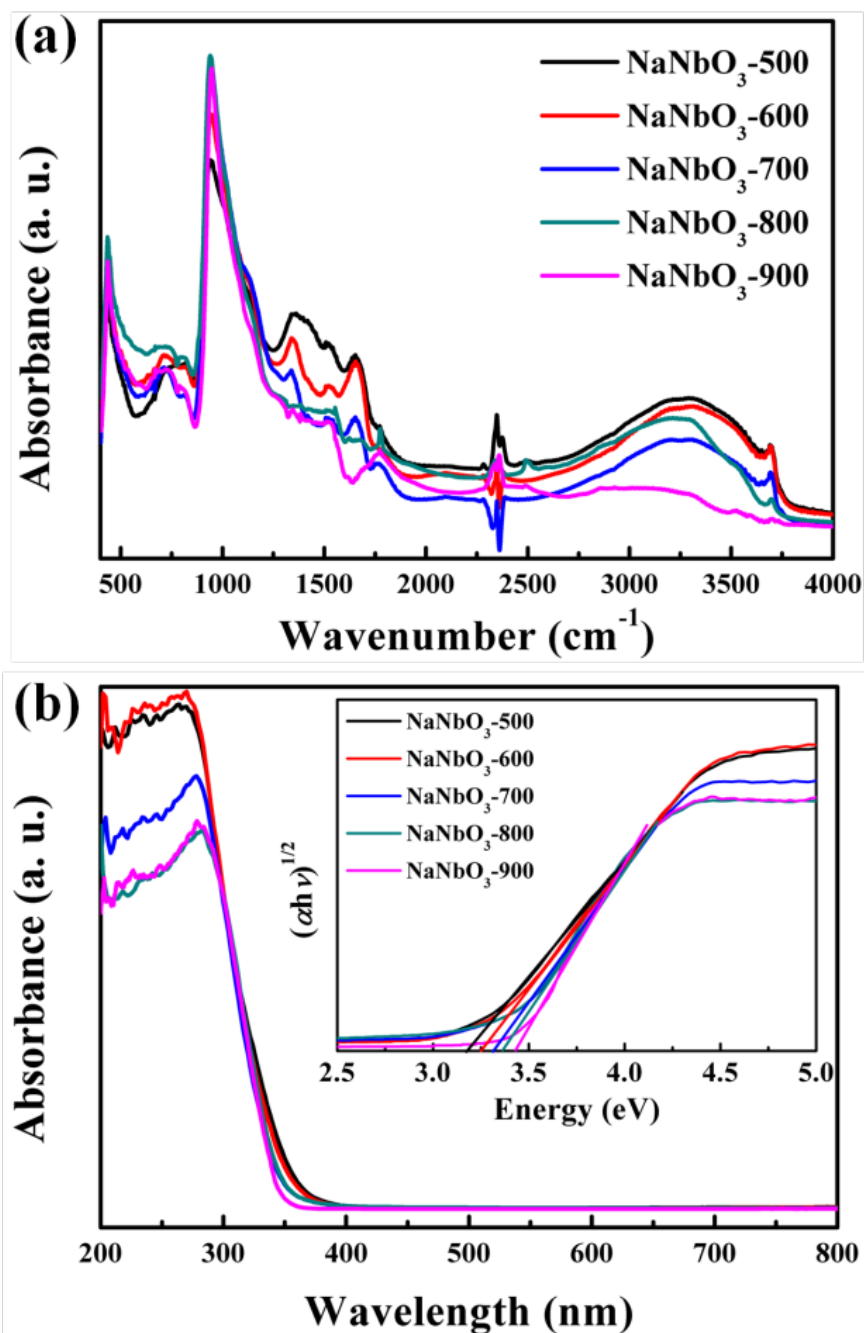
**Fig. S6** TEM (a) and HR-TEM (b) images of NaNbO<sub>3</sub>-500 sample; TEM images of (c) NaNbO<sub>3</sub>-600, (d) NaNbO<sub>3</sub>-700, (e) NaNbO<sub>3</sub>-800, and (f) NaNbO<sub>3</sub>-900 samples.

When the precursor was oxidized, the particle size has an evident raise along with the evolution of the oxidation temperature. The HR-TEM of NaNbO<sub>3</sub>-500 gives the clear evidence that, the particle is crystallized in single crystal and exhibits two mutually perpendicular fringes with the



same *d*-spacing, 3.90 Å, which could be indexed to the {100} and {010} plane of cubic NaNbO<sub>3</sub>. From the XRD patterns, the phase transition between the samples oxidized at 600 and 700°C offers a rapid crystal reform and results in a much larger particle size when the sample was annealing at 700°C. In the view of TEM images, the oxidation temperature could affect the crystal growth and a lower oxidation temperature will encourage the smaller NaNbO<sub>3</sub> particles. The BET surface areas were measured as 42.1, 27.5, 12.4, 8.9, and 6.8 m<sup>2</sup>·g<sup>-1</sup> for the samples oxidized from 500 to 900°C, respectively.

**SI-7 The FT-IR and UV-Vis absorption spectra of the oxidation temperature relative NaNbO<sub>3</sub> samples.**



**Fig. S7** (a) FT-IR absorption spectra and (b) UV-Vis absorption spectra of the as-prepared oxidation temperature relative NaNbO<sub>3</sub> samples. The inset in figure (b) shows the corresponding  $(\alpha h\nu)^{1/2} \sim h\nu$  curves.

When the oxidation temperature is changed from 500°C to 600°C, the intensities of these peaks ranging from 1400 to 1700  $\text{cm}^{-1}$  which are recognized as the vibrations of C-O bond of carbonate ions decrease. If the temperature continually increases to higher than 700°C, the signals of carbonate ions nearly disappeared. The higher oxidation temperature removed the carbonate ions that are coordinated on the  $\text{NaNbO}_3$  surface. The decreasing amount of surface carbonate ions induces that the surface energy of  $\text{NaNbO}_3$  increases and the crystal structure of  $\text{NaNbO}_3$  reforms from cubic to orthorhombic phase with the annealing energy. UV-visible absorption spectra of the  $\text{NaNbO}_3$  powder samples oxidized at different temperatures are shown as Fig. S5(b). All the samples have the similar intense absorption with sharp edges in the UV region. From the  $(\alpha h\nu)^{1/2} \sim \text{Energy}$  plot, the values of the band gaps for as-prepared  $\text{NaNbO}_3$  samples are determined to be 3.17 3.25 3.32, 3.37 and 3.44 eV, respectively. This band-gap extending from low to high oxidation temperature is consistent with the phase transition from cubic to orthorhombic phase.