SUPPLEMENTARY TABLE & FIGURES

Supplementary Table 1. Material properties of CdS, ZnS, and calculated Cd_{0.5}Zn_{0.5}S^{1,2}

| Material | CdS | Calculated Cd _{0.5} Zn _{0.5} S | ZnS |
|--------------------------|--|--|-------------|
| Crystal structure | wurtzite | hexagonal | wurtzite |
| Lattice constant (a / c) | 4.14 / 6.71 | 3.98 / 6.485 | 3.82 / 6.26 |
| Band gap | 2.37 eV | 2.85 eV | 3.56 eV |
| Interfacial strain | $\frac{3.82-3.98}{3.98}\times 100\% = -4.02\%$ at the interface between Cd $_{0.5}$ Zn $_{0.5}$ S core QD and ZnS shell QD | | |

The emitted 442-nm wavelength blue light would be produced via the exciton recombination process at the $Cd_{0.5}Zn_{0.5}S$ core QD and the red shift induced by the compressive strain at the interface between the $Cd_{0.5}Zn_{0.5}S$ core QD and ZnS shell QD. In order to understand this, we estimated the interfacial strain depending on core QD and shell QD materials. The interfacial strain of $Cd_{0.5}Zn_{0.5}S/ZnS$ core/shell QD was calculated by using the references presenting the lattice constants of the $Cd_{0.5}Zn_{0.5}S$ core and ZnS shell QD, indicating a compressive strain of \sim 4.02%, as shown in Table 1. In addition, the energy band-gap of the $Cd_{0.5}Zn_{0.5}S$ core QD was calculated by using the Vegard's law, given by^{3, 4}

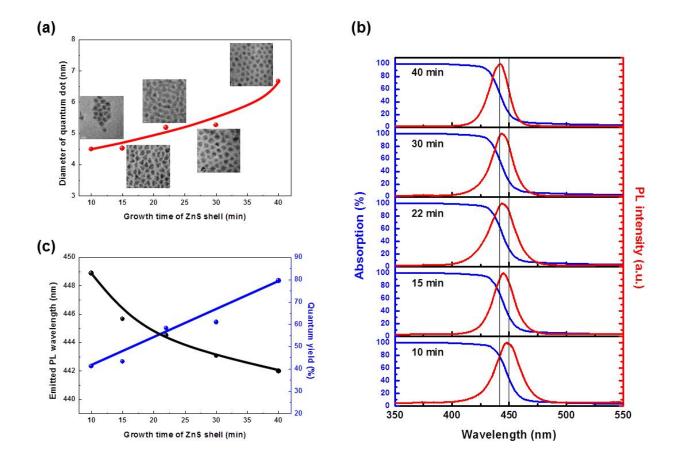
$$\alpha_{Cd_{x}Zn_{1-x}S} = x\alpha_{CdS} + (1-x)\alpha_{ZnS}$$

$$E_{g,Cd_{x}Zn_{1-x}S} = xE_{g,CdS} + (1-x)E_{g,CdS} - bx(1-x)$$

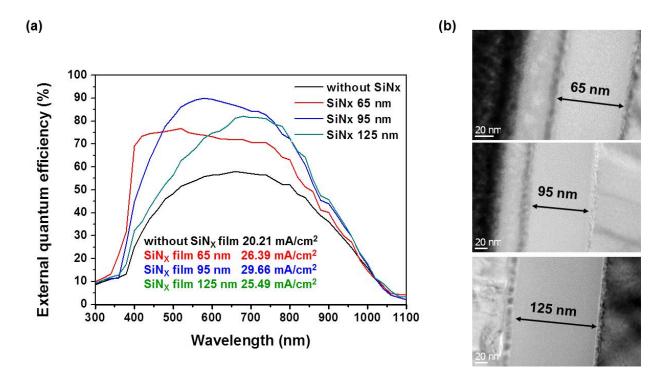
where ${}^{\alpha}{}_{Cd_{x}Zn_{1-x}S}$, ${}^{\alpha}{}_{CdS}$, ${}^{\alpha}{}_{ZnS}$ are the lattice parameters of $Cd_{x}Zn_{1-x}S$, CdS, ZnS and ${}^{E}{}_{g,Cd_{x}Zn_{1-x}S}$, $E_{g,CdS}$, ${}^{E}{}_{g,ZnS}$ are band gap of $Cd_{x}Zn_{1-x}S$, CdS, ZnS and b is the band gap bowing parameter of the $Cd_{x}Zn_{1-x}S$. The calculated band gap of the $Cd_{0.5}Zn_{0.5}S$ core QD is 2.85 eV at 0.45 eV of band gap bowing parameter, which will emits a PL peak at 435-nm in wavelength. However, the measured

PL (emitted blue light) of the $Cd_{0.5}Zn_{0.5}S/ZnS$ core/shell QD peaked at 442-nm in wavelength, corresponding to ~2.81eV. This wavelength difference (a red shift of ~ 7nm) would be originated from the interfacial compressive strain. Therefore, the 442-nm wavelength blue light was emitted by the red shift induced by the interfacial compressive strain from the electron-hole recombination at the energy band gap of the $Cd_{0.5}Zn_{0.5}S$ core QD (2.85 eV).

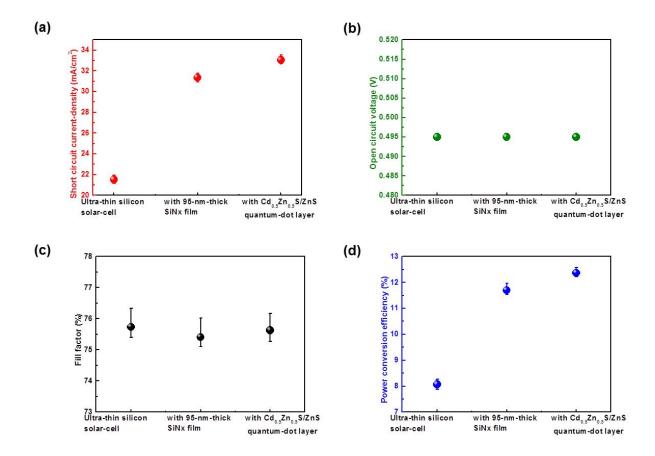
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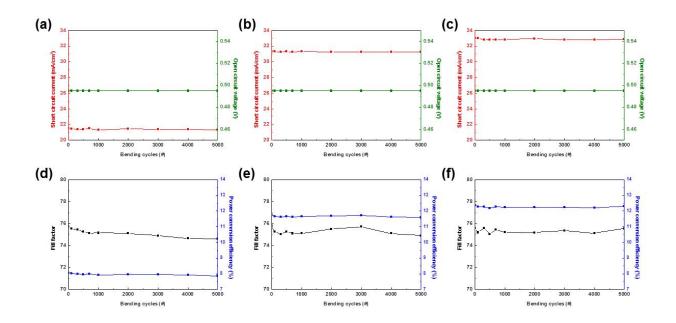
Supplementary Figure 1. Dependency of optical properties on the growth time of the ZnS shell in $Cd_{1-X}Zn_XS/ZnS$ core/shell QDs. (a) Diameter of $Cd_{1-X}Zn_XS/ZnS$ core/shell QDs calculated by observing TEM images, (b) Absorption spectrums and PL peaks, and (c) Emitted PL wavelength and quantum yield.



Supplementary Figure 2. Effect of the SiN_X anti-reflective film thickness on EQE and J_{SC} . (a) EQE as a function of wavelength and (b) TEM images as a function of the SiN_X anti-reflective film thickness.



Supplementary Figure 3. Photovoltaic performance for the flexible ultra-thin silicon solar-cells, flexible ultra-thin silicon solar-cells with SiN_X anti-reflective film, and flexible ultra-thin silicon solar-cells with SiN_X reflective film and $Cd_{0.5}Zn_{0.5}S/ZnS$ core/shell QDs. (a) J_{SC} , (b) V_{OC} ,(c) FF, and (d) PCE by the average (point), maximum (higher bar) and minimum (lower bar) among six samples



Supplementary Figure 4. Bending fatigue performance for flexible ultra-thin silicon solar-cells, flexible ultra-thin silicon solar-cells with SiN_X anti-reflective film, and flexible ultra-thin silicon solar-cells with SiN_X anti-reflective film and $Cd_{0.5}Zn_{0.5}S/ZnS$ core/shell QDs. (a), (b) and (c) J_{SC} and V_{OC} , (d),(e), and (f) FF and PCE as a function of bending cycles.